



2D-correlation analysis applied to *in situ* and *operando* Mössbauer spectroscopy

Laurent Aldon*, Alexis Perea

ICGM/AIME (UMR 5253 CNRS), CC 15-02, Université Montpellier II, Place E. Bataillon, 34095 Montpellier Cedex 5, France

ARTICLE INFO

Article history:

Received 28 May 2010

Received in revised form 29 July 2010

Accepted 10 August 2010

Available online 17 August 2010

Keywords:

2D-correlation spectroscopy

Mössbauer spectroscopy

Lithium battery

Olivine

Data treatment

ABSTRACT

In this paper we propose a new way for Mössbauer data treatment when numerous experimental spectra are recorded in *operando* conditions depending on a perturbation. In our example, the perturbation is the Li amount extracted from a positive electrode material $\text{LiFe}_{0.75}\text{Mn}_{0.25}\text{PO}_4$. In other cases perturbation could be the recording temperature, the pressure or kinetic parameter or even time for isothermal experiments. From analysis of both synchronous and asynchronous 2D-correlation spectra, we can focus our attention on the intensity variations at some specific positions deduced from cross-peaks. This analytical method is very powerful when overlap between absorption lines is observed. This is typically the case when $\text{Fe}^{2+}/\text{Fe}^{3+}$ contributions are simultaneously present in a Mössbauer spectrum at lower velocities.

© 2010 Elsevier B.V. All rights reserved.

1. Introduction

The electrochemical reactions of battery materials have been extensively studied by various electrochemical and spectroscopic techniques [1,2]. We recently determined Lamb–Mössbauer factors of both FePO_4 and LiFePO_4 compounds [3] in order to get precise contributions of each formed/reacted species during electrochemical process. Mössbauer is a powerful technique to follow electrochemical reactions studied by *ex situ* [4,5], *in situ* [6] or *operando* conditions and also for characterization of pristine materials. In some case it is possible to enriched electrode materials with a Mössbauer isotope (^{57}Fe , ^{119}Sn) used as a local probe in $\text{Li}_4\text{Ti}_5\text{O}_{12}$ [7–9], $\text{Li}_2\text{Ti}_3\text{O}_7$ [10,11] for instance. In other cases spectra are broadened due to a mixture of phases or an amorphization of the particles. The help of electronic structure calculation [12,13] allows non-trivial experimental attributions when mathematical ambiguities due to overlapping peaks was observed. Interpretation of both isomer shift and quadrupole splitting variations was explained with such theoretical approaches [14,15]. In the case of glassy compounds quadrupole splitting distributions have been used [16,17] based on previous works [18–20].

Rechargeable lithium batteries have proven to be one of the most successful solutions to achieve good electrochemical performances in terms of cycleability, low cost, environmental benefit, easy to produce and safety in handling and operation. Phosphate materials ($\text{Li}_3\text{Fe}_2(\text{PO}_4)_3$ [21]) have been investigated in the past

for their good ionic properties as solid electrolytes. The LiFePO_4 compound has been chosen as cathode material because of the requirements [22,23]. Pure LiFePO_4 (triphylite) and FePO_4 (heterosite, berlinite [24,25], high pressure [26] and amorphous [27] polymorphs) have been earlier studied for their electrochemical performances [28–30] and their physico-chemical properties like antiferromagnetism [31] by X-ray [32,33] and neutron [34,35] diffraction techniques, Mössbauer spectroscopy [36–38], Raman [39,40] and infra-red [41–43] spectroscopies, X-ray absorption [44] and X-ray photoelectron spectroscopies [45,46]. The origin of capacity loss during the first cycle was also discussed for this good candidate as cathode material [47]. Due to low ionic and electronic conductivity for LiFePO_4 , and to its ageing in presence of moisture [48,49], optimized particle size [50], mineral coating or substitutions ($\text{LiFe}_{1-x}\text{Mn}_x\text{PO}_4$ [51–54] or $\text{Li}_{1-x}\text{Fe}_{1+x}\text{P}_{1-x}\text{Si}_x\text{O}_4$ [55]), doping by transition metals [56], or carbon composites [57] have been proposed to overcome these issues.

In this paper we illustrate, taking advantage of 2D-correlation spectroscopy analysis, a way to analyse Mössbauer spectra recorded in *operando* conditions. We have used Python [58] programming language with Numpy [59] and Pylab [60] libraries to get 2D-correlation spectra. This concept of 2D-correlation was initially proposed by Noda [61,62]. In general 2D-correlation spectra are obtained using a perturbation (temperature [63,64], pH [65], lithium amount [66], kinetic data [67]) applied to the pristine material investigated by a given or several experimental spectroscopies. More details on this analytical method can be found here [68]. From our knowledge 2D-correlation spectroscopy analysis (2D-COSA) used for various experimental techniques (infra-red spectroscopy [69,70], mass spectral data [71], resonance light scattering [72],

* Corresponding author. Tel.: +33 467143354; fax: +33 467143862.
E-mail address: laurent.aldon@um2.fr (L. Aldon).

X-ray photoelectron spectroscopy [73]) have not yet used for Mössbauer spectroscopy. We present here an application of 2D-COSA for ^{57}Fe *in situ* Mössbauer spectroscopy during electrochemical extraction of lithium from $\text{LiFe}_{0.75}\text{Mn}_{0.25}\text{PO}_4$ with olivine structure.

2. Theoretical basis

2.1. Generalized 2D-correlation spectroscopy

A set of m experimental Mössbauer spectra are recorded as a function of a perturbation variable x (e.g., Li amount extracted/inserted during electrochemical tests) ranging from x_{\min} to x_{\max} . The Li variation between two collected spectra is $\Delta x = (x_{\max} - x_{\min}) / (m - 1)$. This continuous variation induces a broadening of the experimental spectra. Each raw spectrum is folded and normalized giving a set of m perturbation-induced dynamic spectra, $y(v, x)$ is then transformed into a set of 2D-correlation spectra by a form of cross-correlation analysis. We first define the reference spectrum $\bar{y}(v)$ of the system to be the averaged spectrum given by:

$$\bar{y}(v) = \frac{1}{m} \sum_{j=1}^m y(v, x) \quad (1)$$

Then we need the dynamic spectra obtained by:

$$\tilde{y}(v, x) = y(v, x) - \bar{y}(v) \quad (2)$$

The intensity of 2D-correlation spectrum $X(v_1, v_2)$ represents the quantitative measure of a comparative similarity or dissimilarity of spectral intensity variations measured at two different velocities, v_1 and v_2 , along with a fixed interval. In a Mössbauer spectrum v_i corresponds to the position of a Lorentzian absorption line. In order to simplify numerical manipulation, $X(v_1, v_2)$ is treated as a complex number:

$$X(v_1, v_2) = S(v_1, v_2) + iA(v_1, v_2) \quad (3)$$

comprising two orthogonal components, known respectively as the synchronous $S(v_1, v_2)$ and the asynchronous $A(v_1, v_2)$ 2D-correlation intensities.

Using Python programming language, $S(v_1, v_2)$ is calculated as follows:

$$S = \text{dot}(y_dyn.\text{conj}().\text{transpose}(), y_dyn) / \text{float}(m - 1)$$

The intensity of the synchronous 2D-correlation spectrum represents the simultaneous changes of two separate spectral intensity variations measured at v_1 and v_2 within the interval between x_{\min} and x_{\max} . The autocorrelation spectrum $a(v_i)$, defined by the diagonal of S matrix, presents peaks called autopeaks, whose magnitude is always positive. It represents the overall extent of spectral intensity variation at the specific velocity v_i within the interval between x_{\min} and x_{\max} . Cross-peaks located at the off-diagonal positions of a synchronous 2D-spectrum represent simultaneous changes of spectral intensities observed at two different velocities v_1 and v_2 . Such a synchronized change suggests the possible existence of a coupled origin of the spectral intensity variations. The sign of synchronous cross-peak becomes positive if the spectral intensities of the two velocities corresponding to the coordinates of the cross-peak are either increasing or decreasing together within the observation interval. However, the negative sign of cross-peaks indicates that one of the spectral intensities is increasing while the other is decreasing.

The asynchronous 2D-correlation spectrum $A(v_1, v_2)$ is calculated in Python as follows:

$$A = \text{dot}(y_dyn.\text{conj}().\text{transpose}(), \text{dot}(N_{jk}, \text{spectres_dyn})) / \text{float}(m - 1)$$

In this previous expression N_{jk} is the Hilbert–Noda matrix calculated as:

$$N_{jk} = \frac{1}{\pi(k - j)} \quad \text{or} \quad 0 \quad \text{if} \quad j = k$$

The intensity of an asynchronous 2D-correlation spectrum represents sequential, but not coincidental, changes of spectral intensities measured separately at v_1 and v_2 . Cross-peaks develop only if the intensity varies out of phase with each other. The sign of asynchronous cross-peak provide useful information on sequential order of events observed by the Mössbauer spectroscopy along the lithium content. If the signs of synchronous and asynchronous cross-peaks are the same, the intensity change at v_1 occurs before v_2 . If the signs of synchronous and asynchronous cross-peaks are different, the intensity change at v_1 occurs after v_2 .

3. Experimental

3.1. Synthesis procedure

$\text{LiFe}_{0.75}\text{Mn}_{0.25}\text{PO}_4$ pristine compound was obtained by solid-state reaction. The precursors: Li_2CO_3 (Sigma–Aldrich, 99%), $\text{FeC}_2\text{O}_4 \cdot 2\text{H}_2\text{O}$ (Labosi, 99.99%), $\text{Mn}^{\text{II}}(\text{CH}_3\text{COO})_2$ (Sigma–Aldrich, 98%) and $\text{NH}_4\text{H}_2\text{PO}_4$ (Acros Organics, 99.99%) were first ball milled during 1 h 30 min and then thermally treated in a tube furnace under argon flow for 8 h at 800 °C.

3.2. X-ray diffraction pattern

$\text{LiFe}_{0.75}\text{Mn}_{0.25}\text{PO}_4$ pristine compound has been characterized by X-ray power diffraction with a Philips $\theta - 2\theta$ diffractometer using Cu K α radiation ($\lambda = 1.5418 \text{ \AA}$) and a nickel filter.

Crystallographic reference data for LiFePO_4 (triphylite) has been taken from power diffraction file (JCPDS #81–1173, $a = 10.332 \text{ \AA}$, $b = 6.010 \text{ \AA}$, $c = 4.692 \text{ \AA}$). These values and the orthorhombic Pnma space group have been used in the initial stages of refinement of the cell parameters for our $\text{LiFe}_{0.75}\text{Mn}_{0.25}\text{PO}_4$ compound.

3.3. ^{57}Fe Mössbauer measurements

Mössbauer spectra have been recorded in the constant acceleration mode and in transmission geometry on a standard Mössbauer spectrometer composed of devices from Ortec and WissEl. A $^{57}\text{Co}(\text{Rh})$ source with a nominal activity of 10 mCi have been used. Each Mössbauer spectrum has been collected every 128 min with a deintercalation uptake $\Delta x = 0.053 \text{ Li}$ corresponding to an electrochemical rate of C/40 (1 Li extracted in 40 h). The source has been always kept at RT. Calibration of the velocity scale was done using absorption lines of a 25 μm iron foil. Isomer shifts are given relative to $\alpha\text{-Fe}$.

3.4. Electrochemical measurements

For Mössbauer measurements in *operando* conditions, we used a designed cell allowing to use it in a transmission configuration. The cell has been designed in such a way that gamma rays can go through the Be windows which are the current collector, the sample under study is in contact with an aluminum foil avoiding corrosion of the Be windows at high potential, and the lithium foil used as reference electrode. Olivine has been used as cathode material with carbon black for better electronic conductivity (80:20) vs. Li used as anode with a Whatman paper as separator using 1 M LiPF_6 in EC:PC:DMC. We used a rate of C/40, corresponding to 1 Li extracted in 40 h with a potential window ranging from 3.2 V to 4.3 V.

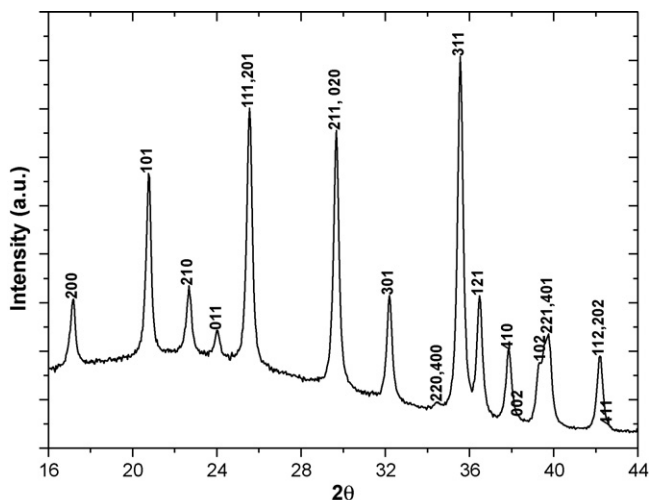


Fig. 1. X-ray diffraction pattern of pristine $\text{LiFe}_{0.75}\text{Mn}_{0.25}\text{PO}_4$. Cell parameters determined by LeBail method [74] using the Pnma space group are $a=10.356(6)\text{Å}$, $b=6.025(4)\text{Å}$, $c=4.703(3)\text{Å}$.

4. Results and discussion

For $\text{LiFe}_{0.75}\text{Mn}_{0.25}\text{PO}_4$ sample, XRD measurements suggest pure phase within the detection limit of the apparatus as shown in Fig. 1. Our pristine material present XRD lines indexed in the Pnma space group with a set of cell parameters $a=10.356(6)\text{Å}$, $b=6.025(4)\text{Å}$ and $c=4.703(3)\text{Å}$. Refined cell parameters have been obtained by pattern matching using FullProf program [74]. These values are in agreement with those published earlier [75].

In Fig. 2 we have represented the ^{57}Fe Mössbauer spectrum of the pristine material. The main contribution (91%) with an isomer shift (IS) of $1.20(1)\text{mm s}^{-1}$ and a quadrupole splitting (QS) of $2.96(1)\text{mm s}^{-1}$ correspond to Fe^{2+} in the olivine structure. These values are in agreement with those found in the literature [37]. Concerning the Fe^{3+} contribution detected in our Mössbauer spectra, the hyperfine parameters are $\text{IS}=0.31(4)\text{mm s}^{-1}$ and $\text{QS}=0.85(8)\text{mm s}^{-1}$. These values are not very closed to those of $\text{Li}_x\text{FePO}_4(\text{OH})_x$ (taverite) [55] the reaction product obtained from LiFePO_4 ageing under moisture. The presence of such an impurity is not excluded since about 9% of Fe^{3+} is present and only 90% of

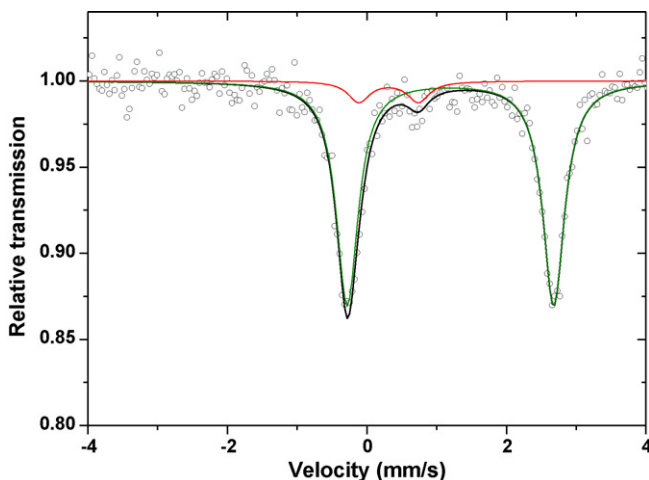


Fig. 2. ^{57}Fe Mössbauer spectrum recorded at room temperature. Experimental (open circle) and calculated spectrum (solid line) are represented. A small contribution (in red) due to Fe^{3+} impurity (see text for explanation) is also shown for clarity. (For interpretation of the references to color in this figure legend, the reader is referred to the web version of the article.)

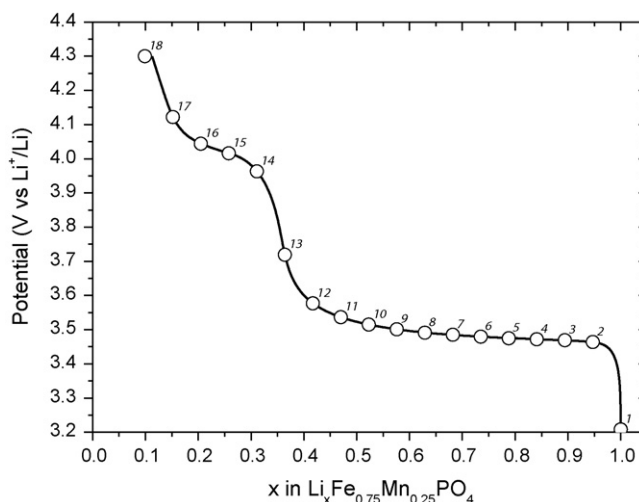


Fig. 3. Electrochemical curve recorded during lithium extraction from $\text{LiFe}_{0.75}\text{Mn}_{0.25}\text{PO}_4$. The numbered labels correspond to the file number of the recorded ^{57}Fe Mössbauer spectra in *operando* conditions.

the Li can be extracted by electrochemical reaction as can be seen in Fig. 3. An other explanation of the Fe^{3+} contribution may due to Fe_2O_3 . From Mössbauer data recorded at various temperatures, it has been shown that anisotropy constant can be determined [76]. Hence for $\alpha\text{-Fe}_2\text{O}_3$ ^{57}Fe Mössbauer spectra present rather a doublet instead of a sextet at room temperature when particles are nano-sized. This doublet presents hyperfine parameters ranging from $\text{IS}=0.32\text{mm s}^{-1}$ $\text{QS}=0.98\text{mm s}^{-1}$ ($D=4\text{nm}$) to $\text{IS}=0.38\text{mm s}^{-1}$, $\text{QS}=0.44\text{mm s}^{-1}$ ($D=18\text{nm}$) depending on the mean particle size. So, in our case, both IS and QS could be some Fe_2O_3 formed at high temperature. From linear correlation between QS and nanoparticle size we found a mean particle size of $D=7\pm 2\text{nm}$ for our Fe_2O_3 impurity. Recently, Rho et al. have mentioned [77] the formation of both Fe_2P (singlet+doublet) and FeP ($\text{IS}=0.6(3)\text{mm s}^{-1}$ and $\text{QS}=0.8(6)\text{mm s}^{-1}$) as impurity in LiFePO_4 .

The electrochemical curve shown in Fig. 3 presents two main regions. Starting for $[x=1\text{Li}, V=3.2\text{V}]$ in $\text{Li}_x\text{Fe}_{0.75}\text{Mn}_{0.25}\text{PO}_4$ we can extract about $\Delta x=0.68\text{Li}$ (Mössbauer spectra #1–13) on a mean voltage of about 3.48 V. This voltage is known to be $\text{Fe}^{2+}/\text{Fe}^{3+}$ redox potential. Then from $[x=0.32\text{Li}, V=3.88\text{V}]$ to $[x=0.10, V=4.3\text{V}]$ we observe a mean voltage at about 4.02 V, corresponding to $\text{Mn}^{2+}/\text{Mn}^{3+}$ redox potential (Mössbauer spectra #14–18). Assuming that 0.91 Li have been used (9% of non-electrochemically active impurity observed in the Mössbauer spectrum, Fig. 2), $\Delta x=0.68\text{Li}$ and $\Delta x=0.23\text{Li}$ are used to oxidize Fe^{2+} to Fe^{3+} and Mn^{2+} to Mn^{3+} , respectively. These values are rather close to the expected one: $0.75 \times 0.91 = 0.683$ to oxidize Fe^{2+} and $0.25 \times 0.91 = 0.227$ to oxidize Mn^{2+} , in a rather good agreement with the stoichiometric ratio $\text{Fe}:\text{Mn}$ of 0.75:0.25.

We have represented in Fig. 4 the ^{57}Fe Mössbauer spectra obtained during electrochemical Li extraction as mentioned above. We clearly see the evolution of the absorption lines upon the perturbation, i.e., the Li content. In spectrum labeled 1, we retrieve the doublet of the olivine ($\text{LiFe}_{0.75}\text{Mn}_{0.25}\text{PO}_4$) constituted of the two peaks located at $\nu_1=-0.28\text{mm s}^{-1}$ and $\nu_2=2.68\text{mm s}^{-1}$, corresponding to the hyperfine parameters $\text{IS}=1.2\text{mm s}^{-1}$ and $\text{QS}=2.96\text{mm s}^{-1}$. After extraction of $\Delta x=0.68\text{Li}$ (spectrum labeled 13), we observe the doublet of the lithiophilite ($\text{Li}_{0.25}\text{Fe}^{3+}_{0.75}\text{Mn}_{0.25}\text{PO}_4$) constituted of the two peaks located at $\nu_1=-0.22\text{mm s}^{-1}$ and $\nu_2=0.98\text{mm s}^{-1}$, corresponding to the hyperfine parameters $\text{IS}=0.38\text{mm s}^{-1}$ and $\text{QS}=1.20\text{mm s}^{-1}$. At the end of lithium extraction (spectrum labeled 18), we observe the doublet of the heterosite ($\text{Fe}^{3+}_{0.75}\text{Mn}^{3+}_{0.25}\text{PO}_4$) constituted of

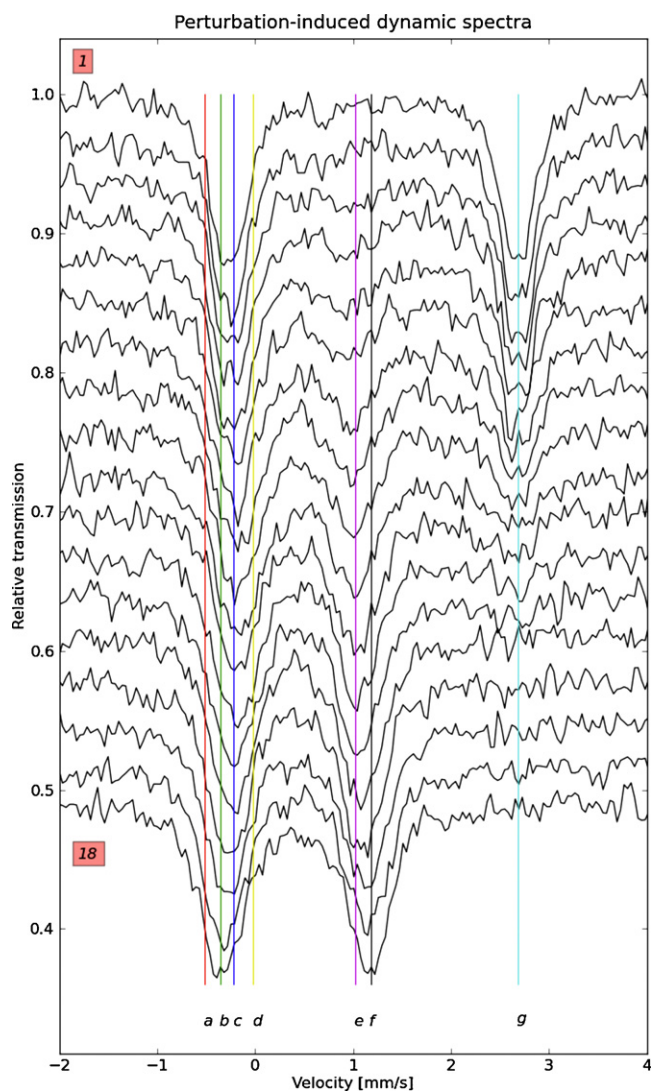


Fig. 4. ^{57}Fe Mössbauer spectra recorded in *operando* conditions during Li extraction from pristine $\text{LiFe}_{0.75}\text{Mn}_{0.25}\text{PO}_4$. Labels 1 and 18 correspond to $\text{LiFe}_{0.75}\text{Mn}_{0.25}\text{PO}_4$ and $\text{Fe}_{0.75}\text{Mn}_{0.25}\text{PO}_4$, respectively. Colored lines [a–g] correspond to cross-correlation peaks positions observed in the synchronous spectra. The evolution of the intensity for those specific positions is discussed in the text. (For interpretation of the references to color in this figure legend, the reader is referred to the web version of the article.)

the two peaks located at $v_1 = -0.36 \text{ mm s}^{-1}$ and $v_2 = 1.18 \text{ mm s}^{-1}$, corresponding to the hyperfine parameters $\text{IS} = 0.41 \text{ mm s}^{-1}$ and $\text{QS} = 1.54 \text{ mm s}^{-1}$.

In Fig. 5 we have plotted the averaged spectrum which give an indication of the regions affected by Li extraction. We see again three main peaks located at about $[-0.2, 1.1 \text{ and } 2.6 \text{ mm s}^{-1}]$. This averaged spectrum is needed to calculate both 2D-correlation synchronous (Fig. 6) and asynchronous (Fig. 7) spectra.

In the 2D-correlation synchronous spectrum (Fig. 6), we see autopeaks with positive contribution (diagonal represented in Fig. 8) and cross-peaks with positive and negative intensities depending on the respective evolution of the intensities of two given velocities. Positive and negative cross-peaks have the following coordinates: ([a, f], [d, e] and [c, g]) and ([a, g], [e, g] and [f, g]), respectively. In Table 1, we have reported the characteristic values of these coordinates.

The transmission intensity is slightly correlated due to the presence of small positive cross-peak with [c, g] coordinates in the 2D-correlation synchronous spectrum. In Fig. 7, cross-peak located

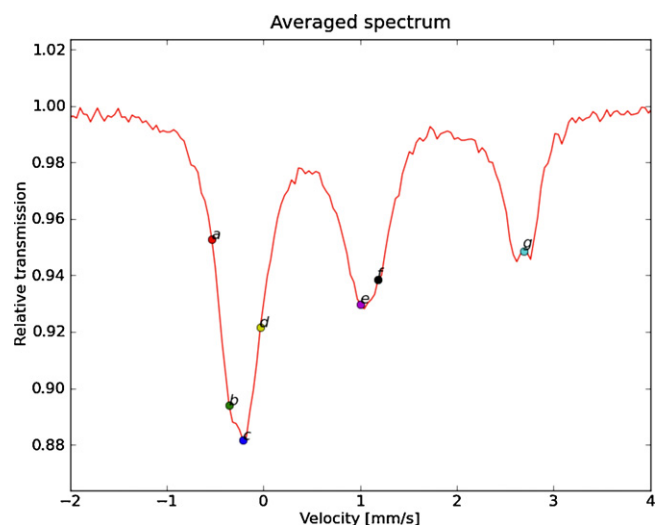


Fig. 5. Averaged spectrum determined from the mean of the 18 Mössbauer spectra recorded during extraction of Li. The main absorption peak located at negative velocity is due to the overlap between contributions of both Fe^{2+} and Fe^{3+} . The peak located near 1 mm s^{-1} correspond to Fe^{3+} contribution and the peak at 2.6 mm s^{-1} correspond to Fe^{2+} contribution.

at [c, g] presents a null intensity in the 2D-correlation asynchronous spectrum. In such analysis, positive (negative) values mean that variation at a given velocity occurs before (after) the perturbation. In the context of Mössbauer spectroscopy, a close-to-zero value in the 2D-correlation asynchronous spectrum suggests that variation observed at two given velocities are coincidental. Hence, these two positions belong to the same doublet of a specie defined by an isomer shift, $\text{IS} = (v_1 + v_2)/2$ and a quadrupole splitting, $\text{QS} = v_2 - v_1$. Using this kind of analysis, we propose an attribution given in Table 1. In Table 2, we have reported the hyperfine parameters deduced from the 2D-correlation analysis and those obtained from a classical fitting procedure. Our 2D-correlation becomes a new tool for analyses of rather complicated Mössbauer spectra presenting some overlaps between absorption lines and small variation from a spectrum to another. Another point to emphasize is that 2D-

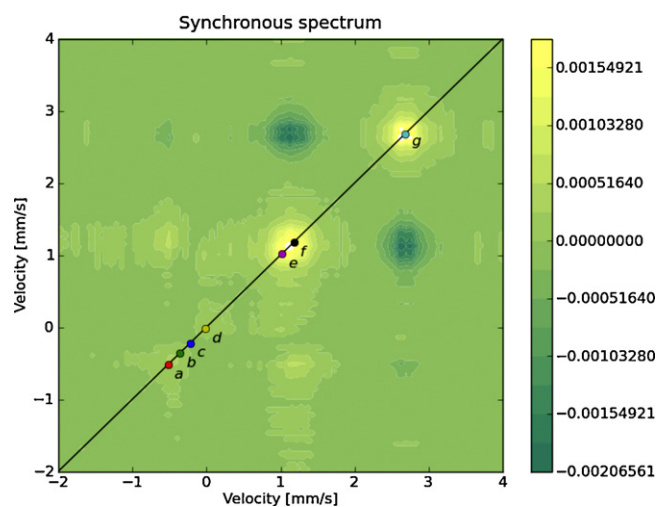


Fig. 6. 2D-correlation synchronous spectrum obtained from experimental ^{57}Fe spectra recorded during electrochemical lithium extraction from $\text{LiFe}_{0.75}\text{Mn}_{0.25}\text{PO}_4$. Positive (light green) and negative (dark green) cross-peaks are observed. Positive cross-peaks have the following coordinates: [a, f], [d, e] and [c, g], while negative cross-peaks have mainly the following coordinates: [a, g], [e, g] and [f, g]. (For interpretation of the references to color in this figure legend, the reader is referred to the web version of the article.)

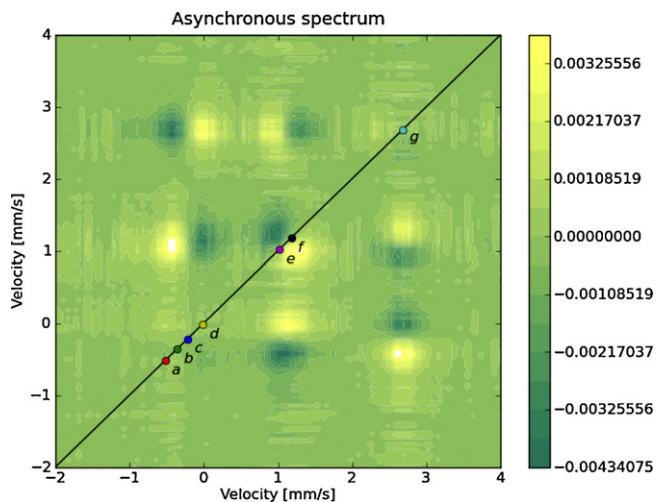


Fig. 7. 2D-correlation asynchronous spectrum obtained from experimental ^{57}Fe spectra recorded during electrochemical lithium extraction from $\text{LiFe}_{0.75}\text{Mn}_{0.25}\text{PO}_4$. Positive cross-peaks (light green) negative cross-peaks (dark green) suggest variation at a given velocity occurs before (after) for another velocity. (For interpretation of the references to color in this figure legend, the reader is referred to the web version of the article.)

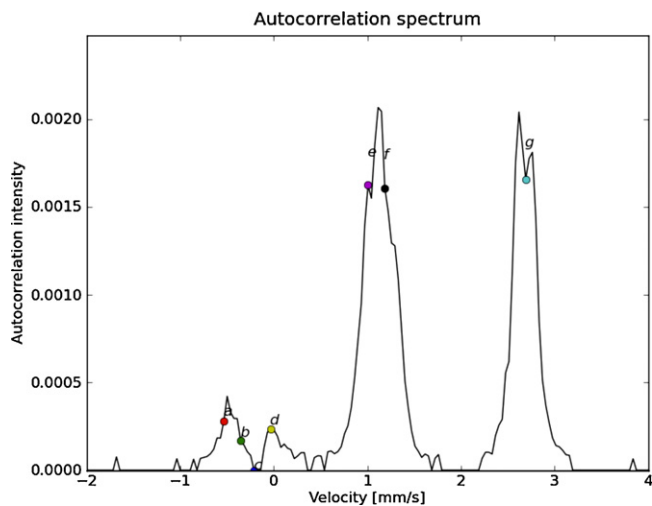


Fig. 8. Autocorrelation spectrum determined from the 2D-correlation synchronous spectrum. Features previously observed in both synchronous and asynchronous correspond to either close to maxima in the autocorrelation spectrum [a, d–f] or minimum [c]. This latter peak is more broadened than those centered at about 2.7 mm s^{-1} . It contains some meaningful information as can be read in the text.

correlation applied to Mössbauer spectroscopy allows to extract both isomer shift and quadrupole splitting of the species using the whole set of spectra instead of using data separately.

The autocorrelation spectrum plotted in Fig. 8 shows the positions of these characteristic values. These sets of coordinates are used to represent the relative transmission variation upon Li content as shown in Fig. 9.

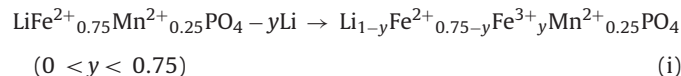
Relative transmission at velocity [g] continuously increases from file #1 to file #13 or file #14 and then still constant up

Table 2

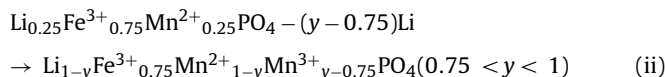
Comparison of isomer shift and quadrupole splitting (δ_{calc} and Δ_{calc} in mm s^{-1}) deduced from 2D-correlation analysis and those obtained by least-square fitting procedure (δ_{fit} and Δ_{fit} in mm s^{-1}).

	ν_1	ν_2	δ_{calc}	Δ_{calc}	δ_{fit}	Δ_{fit}
Fe^{II}	−0.22(c)	2.68(g)	1.23	2.90	1.20	2.96
Fe^{III}_1	−0.02(d)	1.02(e)	0.52	1.04	0.38	1.20
Fe^{III}_2	−0.36(a)	1.18(f)	0.41	1.54	0.41	1.54

to the end of the reaction (file #18). This unambiguous position corresponds to the oxidation process of Fe^{2+} to Fe^{3+} in $\text{Li}_x\text{Fe}_{0.75}\text{Mn}_{0.25}\text{PO}_4$. From $x = 1$ (file #1) to $x = 0.34$ (between file #13 and file #14), we can write the redox process as the following:



Now, focussing our attention on relative transmission at velocities [d–f], we can have an idea of the mechanism involved during the next steps of the Li extraction. In fact, it corresponds in some way to what happens in the Fe^{3+} local environment. It is clear that any variation detected in the Mössbauer spectra may be due to oxidation of Mn^{2+} to Mn^{3+} . Quadrupole splitting is known to be very sensitive to second neighbors effect, and to the charge of the ions. In Fig. 9 we can see that the relative transmission at [e] decreases from file #1 to file #13 and still constant until the end of the Li extraction. While the relative transmission at [d] decreases from file #1 to file #13 and then increases from file #14 to file #18. These observations suggest the following process:



Hence, the presence of Mn^{3+} in the neighborhood of Fe^{3+} induces a higher quadrupole splitting due to the second neighbor shell. This can be related to change induced in the local symmetry of the Fe^{3+}

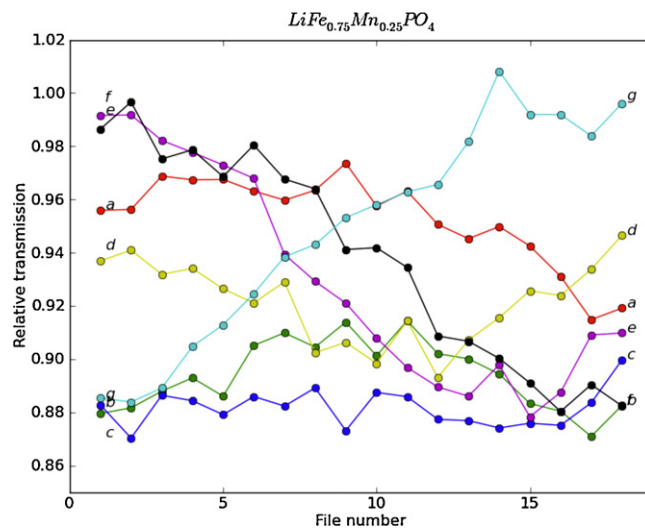


Fig. 9. Evolution of the relative transmission for various velocities [a–g] vs. file number. A decrease in the relative transmission corresponds to an increase of absorption at a given velocity.

Table 1

Velocities of the main features [a–g] deduced from cross-peaks analysis of both 2D-correlation synchronous and asynchronous spectra and autocorrelation spectrum. Some attributions are given as help for the discussion.

Features	a	b	c	d	e	f	g
Velocity (mm s^{-1})	−0.52	−0.36	−0.22	−0.02	1.02	1.18	2.68
Attribution		$\nu_1(\text{Fe}^{\text{III}}_2)$	$\nu_1(\text{Fe}^{\text{II}})$	$\nu_1(\text{Fe}^{\text{III}}_1)$	$\nu_2(\text{Fe}^{\text{III}}_1)$	$\nu_2(\text{Fe}^{\text{III}}_2)$	$\nu_2(\text{Fe}^{\text{II}})$

site due the Jahn–Teller effect on Mn^{3+} ($3d^4$). We also observed this by *in situ* XRD measurement (not the purpose of this paper) and our results are rather in a good agreement with those previously published by Yamada et al. [57].

Concerning the Mössbauer investigations, we have shown by *in situ* and *operando* measurements that ^{57}Fe is directly sensitive to oxidation state of the iron ($\text{Fe}^{2+}/\text{Fe}^{3+}$) during the first stage of the Li extraction as expected in Fig. 3. We have also shown that ^{57}Fe is indirectly sensitive the second neighbor shell through variations of Mn oxidation state. The quadrupole splitting is ranging from 1.20 mm s^{-1} when Fe^{3+} are surrounded by Fe^{3+} and Mn^{2+} sharing the M2 site of the olivine structure, to 1.54 mm s^{-1} when Fe^{3+} are surrounded by Fe^{3+} and Mn^{3+} .

5. Conclusion

2D-correlation spectroscopy applied to Mössbauer spectroscopy during electrochemical studies of electrode materials in real conditions (*in situ* and *operando*) can be considered as a new tool for the Mössbauer community. This analytical technique gives valuable information on the electrochemical mechanisms involved in extraction (insertion) of lithium from (in) electrode materials. Compared to other spectroscopies, Mössbauer spectroscopy is a complementary technique since this electrode material is based on $\text{Fe}^{2+}/\text{Fe}^{3+}$ redox system, ^{57}Fe and gives valuable information at atomic scale in the bulk material, while XPS and Raman are more dedicated to surface analysis. With the help of 2D-correlation analysis, Mössbauer spectroscopy can be used in *operando* conditions, i.e., during Li insertion (extraction) in (from) electrode materials tested in real battery with the presence of current collector (Be windows), separator and Li foil. Hence Mössbauer effect measurements yield the evidence for the oxidation processes taking place in the course of lithium extraction.

2D-correlation technique can be used in other interesting application like catalysis, *in situ* corrosion measurements, f-Lamb–Mössbauer factor determination for instance.

Acknowledgements

The authors thank SAFT Company, Bordeaux and CNRS (contract SAFT/CNRS No. 029888) for the financial support through the Ph.D. grant of Alexis Perea and the Région Languedoc Roussillon for the Mössbauer platform.

References

- [1] L. Aldon, C.M. Ionica, P.E. Lippens, D. Larcher, J.M. Tarascon, J. Olivier-Fourcade, J.-C. Jumas, *Hyperfine Interact.* 167 (2006) 729.
- [2] A. Aboulaich, F. Robert, P.E. Lippens, L. Aldon, J. Olivier-Fourcade, P. Willmann, J.-C. Jumas, *Hyperfine Interact.* 167 (2006) 733.
- [3] L. Aldon, A. Perea, M. Womes, C.M. Ionica-Bousquet, J.-C. Jumas, *J. Solid State Chem.* 183 (2010) 218.
- [4] M. Mouyane, L. Aldon, M. Womes, B. Ducourant, J.-C. Jumas, J. Olivier-Fourcade, *J. Power Sources* 189 (2009) 818.
- [5] A. Ibarra-Palos, C. Darie, O. Proux, J.L. Hazemann, L. Aldon, J.-C. Jumas, M. Morcrette, P. Strobel, *Chem. Mater.* 14 (2002) 1166.
- [6] C.M. Ionica-Bousquet, P.E. Lippens, L. Aldon, J. Olivier-Fourcade, J.-C. Jumas, *Chem. Mater.* 18 (2006) 6442.
- [7] L. Aldon, P. Kubiak, M. Womes, J.C. Jumas, J. Olivier-Fourcade, J.L. Tirado, J.I. Corredor, C. Perez Vicente, *Chem. Mater.* 16 (2004) 5721.
- [8] L. Aldon, P. Kubiak, A. Picard, P.E. Lippens, J. Olivier-Fourcade, J.-C. Jumas, *Hyperfine Interact.* 156/157 (2004) 497.
- [9] P. Kubiak, A. Garcia, M. Womes, L. Aldon, J. Olivier-Fourcade, P.E. Lippens, J.-C. Jumas, *J. Power Sources* 119 (2003) 626.
- [10] M. Van Thournout, L. Aldon, M. Womes, B. Ducourant, J. Olivier-Fourcade, C. Tessier, S. Levasseur, *J. Power Sources* 174 (2007) 1270.
- [11] L. Aldon, M. Van Thournout, P. Strobel, O. Isnard, J. Olivier-Fourcade, J.-C. Jumas, *Solid State Ionics* 177 (2006) 1185.
- [12] P.E. Lippens, L. Aldon, C.M. Ionica, F. Robert, J. Olivier-Fourcade, J.-C. Jumas, *Solid State Ionics* 835 (2005) 249.
- [13] P.E. Lippens, J.-C. Jumas, J. Olivier-Fourcade, L. Aldon, A. Gheorghiu-de la Roque, C. Sénémaud, *J. Phys. Chem. Solids* 61 (2000) 1761.
- [14] M.L. Elidrissi Moubtassim, L. Aldon, P.E. Lippens, J. Olivier-Fourcade, J.-C. Jumas, G. Zegbe, G. Langouche, *J. Alloy Compd.* 228 (1995) 137.
- [15] P.E. Lippens, L. Aldon, J. Olivier-Fourcade, J.-C. Jumas, A. Gheorghiu-de la Roque, C. Sénémaud, *J. Phys. Chem. Solids* 60 (1999) 1745.
- [16] F. Robert, F. Morato, J. Chouvin, L. Aldon, P.E. Lippens, J. Olivier-Fourcade, J.-C. Jumas, B. Simon, Ph. Biensan, *J. Power Sources* 119 (2003) 581.
- [17] L. Aldon, P.E. Lippens, J.-C. Jumas, J. Olivier-Fourcade, H. Bemelmans, G. Langouche, *J. Non-Cryst. Solids* 262 (2000) 244.
- [18] G. Le Caër, J.M. Dubois, *J. Phys. E12* (1979) 1083.
- [19] H.V. Alberto, J.L. Pinto de Cunha, B.O. Mysen, J.M. Gil, N. Ayres de Campos, *J. Non-Cryst. Solids* 194 (1996) 48.
- [20] J.Y. Ping, D.G. Rancourt, Z.M. Stadnik, *Hyperfine Interact.* 69 (1992) 493.
- [21] A.S. Andersson, B. Kalska, P. Eyob, D. Aernout, L. Häggström, J.O. Thomas, *Solid State Ionics* 140 (2001) 63.
- [22] A.K. Padihi, K.S. Nanjundaswamy, J.B. Goodenough, *J. Electrochem. Soc.* 144 (1997) 1188.
- [23] B.L. Ellis, K.T. Lee, L.F. Nazar, *Chem. Mater.* 22 (2010) 691.
- [24] N. Kinomura, M. Shimada, M. Koizumi, S. Kume, *Mater. Res. Bull.* 11 (1976) 457.
- [25] M.P. Pasternak, G.Kh. Rozenberg, A.P. Milner, M. Amanovicz, T. Zhou, U. Schwarz, K. Syassen, R.D. Taylor, M. Hanfland, K. Brister, *Phys. Rev. B* 79 (1997) 4409.
- [26] M.E. Arroyo-de Dompablo, J.M. Gallardo-Amores, M.T. Azcondo, F. Garcia-Alvarado, U. Amador, *J. Phys. Chem. Solids* 67 (2006) 1243.
- [27] S. Okada, T. Yamamoto, Y. Okazaki, J.-I. Yamaki, M. Tokunaga, T. Nishida, *J. Power Sources* 146 (2005) 570.
- [28] T. Shiratushi, S. Okada, J.-I. Yamaki, S. Yamashita, T. Nishida, *J. Power Sources* 173 (2007) 979.
- [29] K. Zaghbi, N. Ravet, M. Gauthier, F. Gendron, A. Mauger, J.B. Goodenough, C.M. Julien, *J. Power Sources* 163 (2006) 560.
- [30] C. Delmas, M. Maccario, L. Croguennec, F. Le Cras, F. Weill, *Nat. Mater.* 7 (2008) 665.
- [31] R.P. Santoro, R.E. Newnham, *Acta Crystallogr.* 22 (1967) 344.
- [32] V.A. Streltsov, E.L. Belokoneva, V.G. Tsirelson, N.K. Hansen, *Acta Crystallogr. B* 49 (1993) 147.
- [33] H.C. Shin, K.Y. Chung, W.S. Min, D.J. Byun, H. Jang, B.W. Cho, *Electrochem. Commun.* 10 (2008) 536.
- [34] G. Rousse, J. Rodriguez-Carvajal, S. Patoux, C. Masquelier, *Chem. Mater.* 15 (2003) 4082.
- [35] C. Delacourt, J. Rodriguez-Carvajal, B. Schmitt, J.M. Tarascon, C. Masquelier, *Solid State Sci.* 7 (2005) 1506.
- [36] M.P. Pasternak, G.Kh. Rozenberg, A.P. Milner, M. Amanovicz, K.E. Brister, R.D. Taylor, *J. Magn. Magn. Mater.* 183 (1998) 185.
- [37] A.S. Andersson, B. Kalska, L. Häggström, J.O. Thomas, *Solid State Ionics* 130 (2000) 41.
- [38] K. Hirose, T. Honma, Y. Doi, Y. Hinatsu, T. Komatsu, *Solid State Commun.* 146 (2008) 273.
- [39] R. Baddour-Hadjean, J.P. Pereira-Ramos, *Chem. Rev.* 110 (2010) 1278.
- [40] C.V. Ramana, A. Mauger, F. Gendron, C.M. Julien, K. Zaghbi, *J. Power Sources* 187 (2009) 555.
- [41] P. Jozwiak, J. Garbarczyk, F. Gendron, A. Mauger, C.M. Julien, *J. Non-Cryst. Solids* 354 (2008) 1915.
- [42] S. Hamelet, P. Gibot, M. Casas-Cabanas, D. Bonnin, C.P. Grey, J. Cabana, J.-B. Leriche, J. Rodriguez-Carvajal, M. Courty, S. Levasseur, P. Carlach, M. Van Thournout, J.M. Tarascon, C. Masquelier, *J. Mater. Chem.* 19 (2009) 3979.
- [43] C.M. Burba, R. Frech, *Electrochem. Acta* 52 (2006) 780.
- [44] A.A.M. Prince, S. Mylswamy, T.S. Chan, R.S. Liu, B. Hanoyer, M. Jean, C.H. Shen, S.M. Huang, J.F. Lee, G.X. Wang, *Solid State Commun.* 132 (2004) 455.
- [45] R. Dedryvère, M. Maccario, L. Croguennec, F. Le Cras, C. Delmas, D. Gonbeau, *Chem. Mater.* 20 (2008) 7164.
- [46] L. El Ouatani, R. Dedryvère, C. Siret, Ph. Biensan, D. Gonbeau, *J. Electrochem. Soc.* 156 (2009) 468.
- [47] A.S. Andersson, J.O. Thomas, *J. Power Sources* 97–98 (2001) 498.
- [48] K. Zaghbi, M. Dontigny, P. Charest, J.F. Labrecque, A. Guerfi, M. Kopec, A. Mauger, F. Gendron, C.M. Julien, *J. Power Sources* 185 (2008) 698.
- [49] M. Cuisinier, J.-F. Martin, N. Dupré, A. Yamada, R. Kanno, D. Guyomard, *Electrochem. Commun.* 12 (2010) 238.
- [50] A. Yamada, S.C. Chung, K. Hinokuma, *J. Electrochem. Soc.* 148 (2001) A224.
- [51] C. Delacourt, C. Wurm, P. Reale, M. Morcrette, C. Masquelier, *Solid State Ionics* 173 (2004) 113.
- [52] C.M. Burba, R. Frech, *J. Power Sources* 172 (2007) 870.
- [53] J. Molenda, W. Ojczyk, K. Swierczek, W. Zajac, F. Frok, J. Dygas, Liu Ru-Shi, *Solid State Ionics* 177 (2006) 2617.
- [54] J.M. Osorio-Guillen, B. Holm, R. Ahuja, B. Johansson, *Solid State Ionics* 167 (2004) 221.
- [55] N. Recham, M. Casas-Cabanas, J. Cabana, C.P. Grey, J.-C. Jumas, L. Dupont, M. Armand, J.M. Tarascon, *Chem. Mater.* 20 (2008) 6798.
- [56] D.X. Gouveia, V. Lemos, J.A.C. de Paiva, A.G. Souza Filho, J. Mendes Filho, S.M. Lala, L.A. Montoro, J.M. Rosolen, *Phys. Rev. B* 72 (2005) 24105.
- [57] A. Yamada, M. Hosoya, S.-C. Chung, Y. Kudo, K. Hinokuma, K.-Y. Liu, Y. Nishi, *J. Power Sources* 119–121 (2003) 232.
- [58] Python (Release 2.6.4 for win32) free download at <http://www.python.org>.
- [59] Numerical Python (Release 1.4.0.dev7335), an Open Source Project developed by D. Ascher, P.F. Dubois, K. Hinsen, J. Hugunin, T. Oliphant.
- [60] Pylab and Matplotlib (Release 0.99.1), a Graphic Library developed by D. Dale, M. Droettboom, E. Firing, J. Hunter.

- [61] I. Noda, J. Mol. Struct. 883–884 (2008) 216.
[62] I. Noda, Appl. Spectrosc. 47 (1993) 1329.
[63] S. Morita, K. Kitagawa, I. Noda, Y. Ozaki, J. Mol. Struct. 883–884 (2008) 181.
[64] H. Ji, S.B. Kim, I. Noda, Y.M. Jung, Spectrochim. Acta A71 (2009) 1873.
[65] X. Dou, B. Yuan, H. Zhao, G. Yin, O. Yukihiro, Sci. China B Chem. 47 (2004) 257.
[66] H.C. Choi, Y.M. Jung, I. Noda, S.B. Kim, J. Phys. Chem. B107 (2003) 5806.
[67] P.J. Tandler, P. de, B. Harrington, H. Richardson, Anal. Chim. Acta 368 (1998) 45.
[68] Y.M. Jung, H.S. Shin, S.B. Kim, I. Noda, Appl. Spectrosc. 56 (2002) 1562.
[69] R. Dluhy, S. Shanmukh, S.-I. Morita, Surf. Interface Anal. 38 (2006) 1481.
[70] D.L. Elmore, R.A. Dluhy, Colloids Surf. A171 (2000) 225.
[71] K.G. Owens, Appl. Spectrosc. Rev. 27 (1992) 1.
[72] J. Yang, X. Chen, R. Fu, Y. Li, W.-A. Luo, M. Zhang, Polym. Test. 28 (2009) 456.
[73] J.-F. Boily, E.S. Ilton, Surf. Sci. 602 (2008) 3637.
[74] J. Rodriguez-Carvajal, Physica B. 192 (1993) 55.
[75] T. Nakamura, K. Sakumoto, M. Okamoto, S. Seki, Y. Kobayashi, T. Takeuchi, M. Tabushi, Y. Yamada, J. Power Sources 174 (2007) 435.
[76] W. Kündig, H. Bömmel, G. Constabaris, R.H. Lindquist, Phys. Rev. 142 (1966) 327.
[77] Y.-H. Rho, L.F. Nazar, L. Perry, D. Ryan, J. Electrochem. Soc. 154 (2007) A283.